



Phase Transformations, Microstructure and Shape Memory Effect of NiTiAg Alloy with Different Atomic Percentages (at. % Ag) Manufactured by Casting Method

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Submitted: 13/09/2020

Accepted: 26/10/2020

Published: 25/04/2021

KEY WORDS

Cast method, DSC, NiTiAg, SMAs, SME, VAR.

ABSTRACT

In this paper, shape memory alloys (SMAs) (NiTi-based) have been manufactured by casting with a different atomic percentage of a silver element (0, 1, 2 and 3 at. % Ag) using a Vacuum Arc Remelting (VAR) furnace. The silver element is added to the binary alloys due to its excellent properties such as (anti-corrosion, anti-bacterial and high electrical conductivity), which make these alloys using in wider applications. These alloys with different atomic percentages (Ni55Ti45Ag0, Ni55Ti44Ag1, Ni55Ti43Ag2 and Ni55Ti42Ag3) have been manufactured. The successful manufacturing process has been achieved and proved via examinations and tests. The FESEM microscopic examinations show that the silver element has been distributed uniformly and homogeneously in the NiTi matrix. Moreover, the emergence of austenite phase, martensite phase and little amount impurities. Regarding the XRD examination, showed that there is an increase in the number of peaks of Ag phase with an increase in the atomic percentage of the silver element, as well to emergence of phase (Ti2Ni) upon heating, phase (Ti002) upon cooling, and phase (Ni4Ti3) is not desired. The starting and finishing of the phase transformations have been determined for all samples by the DSC test. The Shape Memory Effect (SME) for the alloy (Ni50Ti42Ag3) is measured to be about 89.99%.

How to cite this article: S. M. Hussein, K. D., Salman, and A. A. Hussein, "Phase transformations and physical properties of NiTiAg alloy with different atomic percentages (at. % Ag) manufactured by casting method," Engineering and Technology Journal, Vol. 39, Part A, No. 04, pp. 543-551, 2021.

DOI: <https://doi.org/10.30684/etj.v39i4A.1833>

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1. INTRODUCTION

Shape memory alloys (SMAs) are metallic materials (smart materials) that respond to external changes such as temperature and mechanical load, thus producing deformation (generally 2-10%). These materials are described as natural thermo-mechanical actuators that are capable of carrying out several mechanical works [1]. SMAs can even remember their predefined shape after the deformations during the heating process. Nitinol can be martensite phase (B19), austenite phase (B2), or a combination of both phases, depending on the temperature. SMAs usually are transformed between the low-temperature phase called the martensite phase and the high-temperature phase called the austenite phase. Several factors can influence the phase transformations of shape memory alloys such as processing techniques, variation in nickel content, thermal cycling, combination thermomechanical treatment and ternary alloying elements [2]. SMAs (NiTi) that were first produced in 1960 are known for two unique properties; superelastic effect (SE), and shape memory effect (SME), mechanically or thermally activated [3, 4]. NiTi alloy has an excellent shape memory effect property. It is called thermal shape memory alloy because it bends and expands upon cooling, and then returns to its size and shape at a certain temperature, this feature is important for many alloy devices such as (NiTi) [5]. In addition, there are several properties for this alloy such as (corrosion resistance, biocompatibility). This indicates that shape memory alloy (NiTi) has an important role in many important applications such as medical and engineering applications [6].

SMAs (NiTi-based) have been developed depending on application requirements by adding a new element to these binary alloys. Additives have an important role in controlling phase transformation, crystal structure and hysteresis loop widths [7-9]. Such additions that lower the transformation temperatures are (Fe, Mn, Cr, V, Al), while those increase the transformation temperatures are (Au, Pt, Pd, Zr, Hf). The addition of (Nb and Fe) widens the transformation hysteresis whereas (Cu) narrows it. Moreover, these additions have a great influence on mechanical and functional properties [10]. Silver is a promising element because it has important properties and advantages such as (good antibacterial ability, excellent biocompatibility, and thermal stability), in addition to excellent corrosion resistance due to the creation of a layer of oxidation [11, 12]. Adding a silver element to alloys NiTi-based has been presented in [13]. For the first time, the silver element was introduced in the year (1991) using Electron Beam Melting (EBM) technique. The results of this study showed that it is difficult to add silver to the binary alloy due to the high vapor pressure of silver. A vacuum arc remelting (VAR) furnace was used to produce an alloy (NiTiAg) [14, 15]. These studies reported the difficulty of silver element melting due to the high pressure of steam of (Ag). But these studies showed that alloy NiTiAg has strength higher than NiTi as well as improved corrosion rate and biocompatibility. Adding a silver element to the binary alloys NiTi using (VAR) method has been presented in [16]. The results of this approach showed that the prepared alloys are homogeneous and stable. While the silver element was lost with repeated melting due to the high steam pressure. This study showed that the silver element is anti-bacterial (increases the anti-bacterial activity of the alloy NiTi), and rate recovery for alloys manufactured was (72-85%).

The problem of research in the process of manufacturing an alloy (NiTi) with the addition of silver, due to the difficulty of distributing the silver element inside the alloy, as the manufacturing process affects the properties of the produced alloy. In general, there are two main methods of manufacturing SMAs, which are powder metallurgy and casting method. The first method (PM) is not used because in this research an ingot is produced [1,17]. The second method in which the used materials must be of high purity because the purity of the materials affects the homogeneity and thus if the materials are not pure it affects the chemical composition as well as the biological properties. In general, the casting method is classified as Electron Beam Melting (EBM), Vacuum Induction Melting (VIM) and Vacuum Arc Remelting (VAR) [18].

Vacuum Arc Remelting (VAR) does not contain a crucible of graphite, which leads to the production of high-purity alloys (low carbon content) [19], while (VIM) requires a crucible of graphite, which causes impurities in the produced alloys (such as carbon element) and thus the alloy is not of high purity [17]. (EBM) produces high-purity alloys, but during the manufacturing process, some elements evaporate, which leads to a high vacuum pressure during melting [1, 20]. The aim of the present research is to produce a homogeneous alloy with different percentages by using a (VAR) furnace and to study the microstructure, phase transformations.

The rest of the sections of this research consist of the following. The second section consists of the materials used, the manufacturing process and subsequent processes. The third section consists of the results and discussion of all the examinations and tests. The last section consists of the conclusions of this work.

2. EXPERIMENTAL WORK

I. Materials

Nickel (Ni) of high purity (99.2 wt.%), titanium (Ti) of about (99.7 wt.%), while silver (Ag) plates of high purity (99.9 wt.%) used as an additive element at atomic percentages (1, 2 and 3 at. %) to prepare SMAs (Table I).

TABLE I: Different atomic percentages of each (Ni, Ti, Ag).

Samples	Ni (at. %)	Ti (at. %)	Ag (at. %)
1	55	45	0
2	55	44	1
3	55	43	2
4	55	42	3

II. Preparation of the Samples

The chosen elements with the different atomic percentages, shown in Table I, were melted in (VAR) furnace type (ALD Vacuum Technologies AG) at 1350°C without graphite crucible. Figure (1.a) shows the VAR furnace using in this work to prepare shape memory alloys with the different atomic percentages of Ag. In this work, there are two VAR methods used to manufacture shape memory alloys. The first method used a consumable electrode made of materials to be melted, while the second method used a non-consumable electrode. Raw metals (Ni, Ti and Ag) are installed on the copper mold and heated by irradiation using argon and the tungsten electrode, when the raw metals are melted down, their shape like the button because of the effect of surface tension. The melting process was followed by solidification (quenching in ice water), and then repeating the melting cycle 4 times to obtain the homogeneity in chemical composition, afterward, all the ingots were annealed at 660°C for 24 h by a programmable electric furnace. After that, the prepared ingots were sectioned by metals cutting machine. After cutting the samples, they were prepared for examinations by X-Ray Diffraction (XRD) analysis, Field Emission Scanning Electron Microscopy (FESEM), physical tests (Differential Scanning Calorimetry (DSC) test and Shape Memory Effect (SME) test) in order to know the starting and finishing of the phase transformations and the rate of shape recovery.

III. Microstructural examinations

A. XRD

XRD analysis was performed for all the samples before and after adding Ag element to define the phases as a result of the casting process. Also, it was used to study the effect of Ag atomic percentage on the resultant phases of the manufactured shape memory alloys. These phases are defined by lab (XRD-6000) System SHIMADZU Europe.

B. FESEM

This FESEM examination was carried out for all the samples before and after adding the silver element at different atomic percentages. Before testing, the samples must be ground by emery papers at 320, 500 and 1000 μm in grit size, and then polished by polishing cloth with chloride paste. After polishing, the samples etched using etching solution (95 mL H₂O + 3.5 mL HNO₃ + 1.5 mL HF) for 10 sec [14]. After sample preparation, the samples coated with gold and palladium spray for 2 min and examined by FESEM type (cam scan Mv 2300).

C. DSC

The samples before and after adding Ag element were tested by Differential Scanning Calorimetry device (SETARAM, model 131 Evo) to estimate the starting and finishing

transformation temperatures (As, Af, Ms, Mf). The weight of the sample of 5-10 g is placed in a small crucible made of aluminum and then heated in a DSC device at 450°C with a heating rate of about 10°C/min. As, Af, Ms and Mf were achieved at -50°C – + 230°C.

D. SME

This test was carried out of the sample containing 3 at. % by pressing the sample about (0.06%) from its original length L_0 [21] using a compression test machine with 1mm/min displacement and then heating the sample in the air. Afterward, the shape memory effect would be calculated by the following formula [22].

$$\text{SME}\% = \frac{L_2 - L_1}{L_0 - L_1} \times 100 \quad (1)$$

Where:

L_0 : Original length of sample (mm).

L_1 : Length of sample after pressing by 0.06%.

L_2 : The return in length of sample after heating.

3. RESULTS AND DISCUSSION

I. XRD analysis

X-ray diffraction was performed by X-Ray diffractometer model (X.R.D-6000) SHIMADZU Europe with $\text{CuK}\alpha$ radiation at the wavelength (λ) = 1.5405 Å for all the samples of SMAs before and after adding Ag element.

Figure 1 shows XRD peaks of SMA NiTi (0 at. % Ag) measured at room temperature at diffraction angles (2θ) ranging about 35° -76.5°. The strongest peaks for the Ni phase created at (2θ) 44.5°, 52° and 76.5° respectively for (111), (200) and (200) planes. While Ti phase created at (2θ) at 35°-43.5° and 63-71° for the planes about (100), (002), (101), (200), (102) and (110) respectively. It must be emphasized that creating Ti_2Ni (Austenite phase) at (2θ) about 35.0°-38.5° and (110) plane. While M (martensitic phase) created at (2θ) about 38.5° and (002) plane.

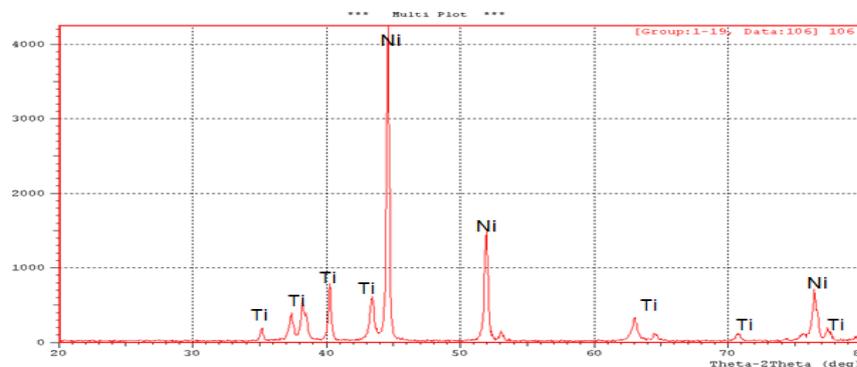


Figure 1: XRD peaks of 0 at. % Ag element.

Figure 2 shows XRD peaks for SMA NiTi with (1 at. % Ag). Here there are three main phases created; Ni, Ti and Ag phases measured at room temperature which are occurred at (2θ) ranging 35°-77.6°. Ni phase is created at (2θ) about 44.5°, 52° and 76.5° for (111), (200) and (111) planes. While Ti phase is created at (2θ) about 35°, 40°, 43.5°, 53°, 63° and 71.0° for (100), (101), (200), (102), (110) and (103) planes respectively. The additive element (Ag) created at (2θ) about 38.5°, 64.6°, 77.6° for (111), (220) and (111) planes respectively.

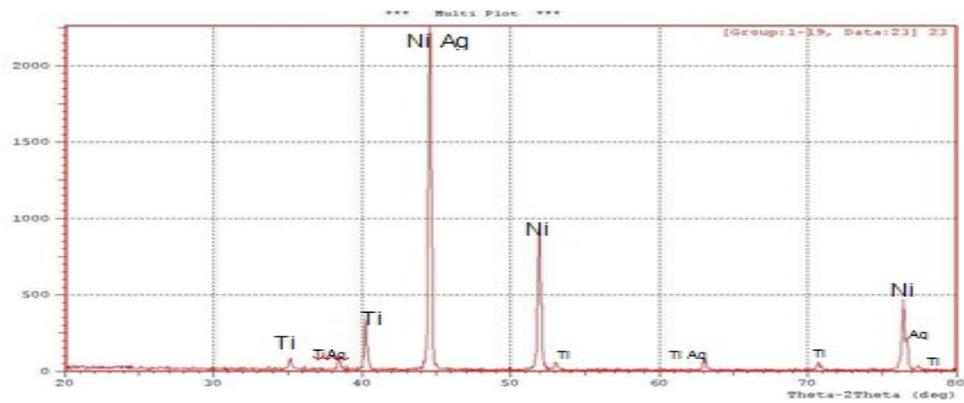


Figure 2: XRD peaks of 1 at. % Ag element.

While Figure 3 shows the XRD peaks for SMAs NiTi with (2 at. % Ag). There are three main phases created at Ni, Ti and Ag phases, these phases occurred at (2θ) ranging about 35° - 77.5° . Ni phase is created at (2θ) about 44.5° , 52° and 76.5° for (111), (200) and (111) planes respectively. While Ti phase is created at (2θ) about 35° , 38.5° , 40.0° , 53° and 71.0° for (100), (002), (101), (102) and (103) planes respectively. Ag phase is created at (2θ) about 37.5° , 43.5° , 44.5° , 63° and 77.5° for (111), (200), (200), (220) and (111) planes respectively.

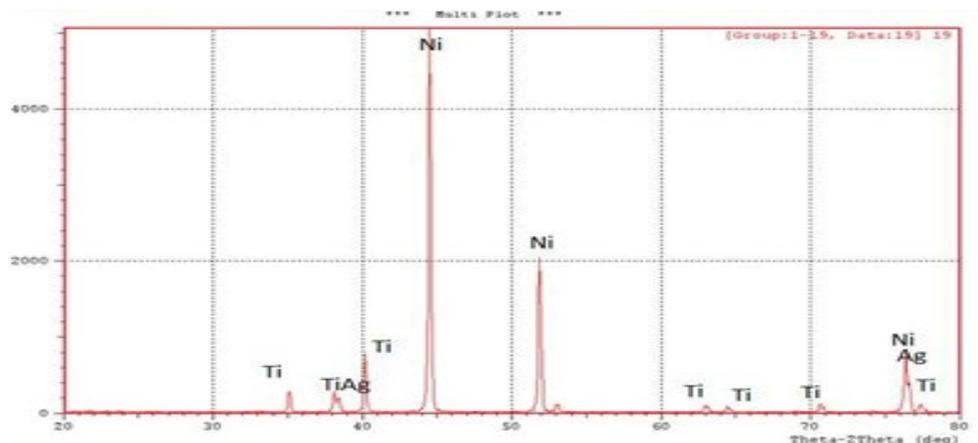


Figure 3: XRD peaks of 2 at. % Ag element.

Finally, Figure 4 shows the XRD peaks for SMAs NiTi with (3 at. % Ag). There are three main phases created as Ni, Ti and Ag phases, these phases occurred at (2θ) ranging about 35° - 77.5° . Ni phase is created at (2θ) about 44.5° and 76.5° for (111), (111) planes respectively. While Ti phase is created at (2θ) about 35° , 38.5° , 40° , 52° and 63° for (100), (002), (101), (200) and (110) planes respectively. Ag phase is created at (2θ) about 37.5° , 43.5° , 44.5° , 53° , 62° , 71° and 77.5° for (111), (200), (200), (102), (220), (103) and (111) planes respectively.

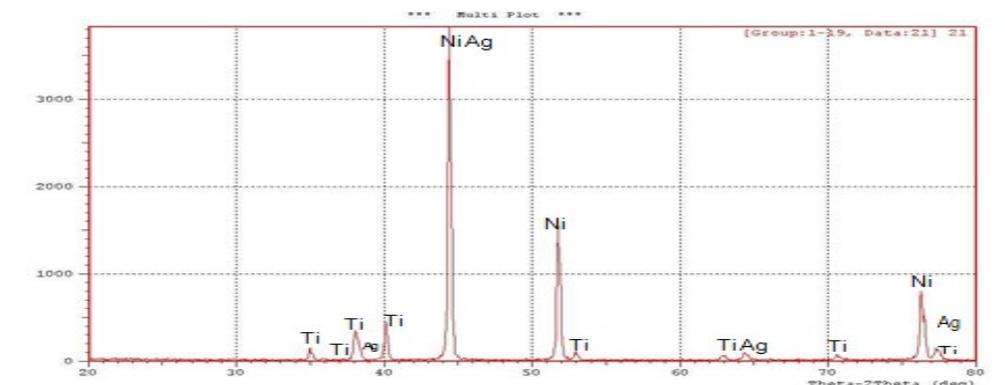


Figure 4: XRD peaks of 3 at. % Ag element.

From the results of XRD analysis, it is shown that increasing the atomic percentage of Ag leads to an increase in the peaks number of Ag phase

The results of XRD analysis revealed that, in heating, Ti_2Ni (cubic austenite phase) is created at ($2\theta = 37^\circ$) for (110) plane. While, in cooling, there is Ti 002 phase (monoclinic martensite phase) will be created at ($2\theta = 38.5^\circ$) for (002) plane. Also, the results show a creation of undesirable phase Ni_4Ti_3 at (2θ) about 44.5° - 52° for (111) and (200) planes. This phase is created as a result of thermal transformation or the existence of some inclusions formed during melting. This is agreed with [23].

II. FESEM

Figure 5 shows the photomicrographs of SMAs before and after adding Ag element with the different atomic percentage to the NiTi-based alloy. The images reveal the martensite phase (Ti 002) and austenite phase (Ti_2Ni) as a result of thermal transformation with a little number of inclusions which attributed to the impurities associated with the casting process, cutting, grinding and polishing of SMAs prepared in this work. These images agreed with XRD data.

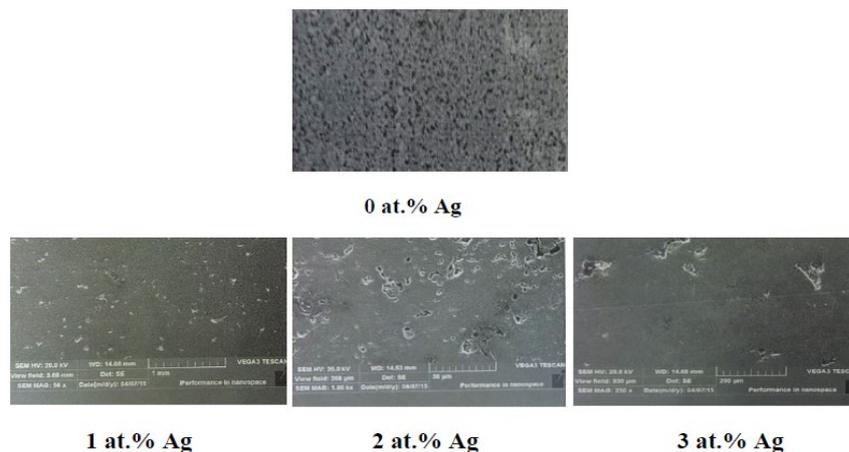


Figure 5: Photomicrographs of the samples for different at. % Ag.

III. DSC

Results of Differential Scanning Calorimeter (DSC) show starting and finishing transformation temperatures, for austenite and martensite phases. They are A_s , A_f , M_s , and M_f respectively, as shown in Table II.

TABLE II: Phase transformation degrees (A_s , A_f , M_s , M_f) for each alloy.

Sample	Transformation Temperature ($^\circ C$)			
	A_s	A_f	M_s	M_f
Ni50Ti45Ag0	84.2	196.3	150	97.4
Ni50Ti44Ag1	-49.2	58.1	149.15	104.9
Ni50Ti43Ag2	83.11	239.3	155.3	102.3
Ni50Ti42Ag3	138.8	179.3	163.01	117.0

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Figures 6, 7, 8 and 9 show the profile of the DSC test for all the samples of SMAs before and after adding the Ag element. These profiles containing two peaks created through heating and cooling which indicated thermal transformation from the austenite phase (B2) to the martensite phase (B19). There are two endothermic peaks formed through the heating process and two exothermic peaks created through the cooling process. This is returned to the formation of the Ni_2Ti phase and fine precipitations which causes the large transformation of B2 to B19 (austenite to the martensite). This is agreed with [24].

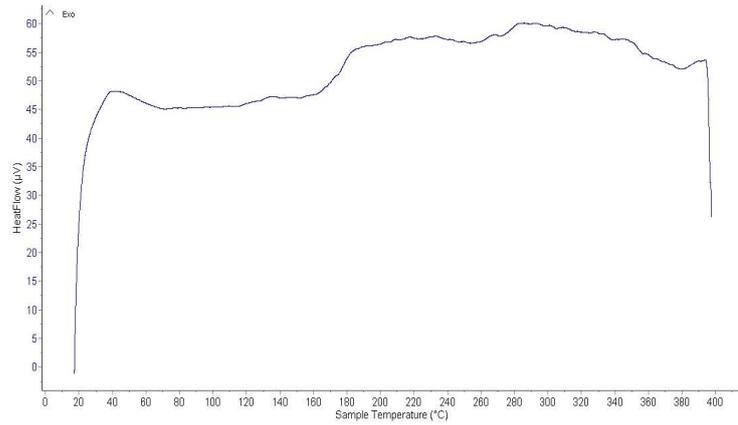


Figure 6: 0% Ag element.

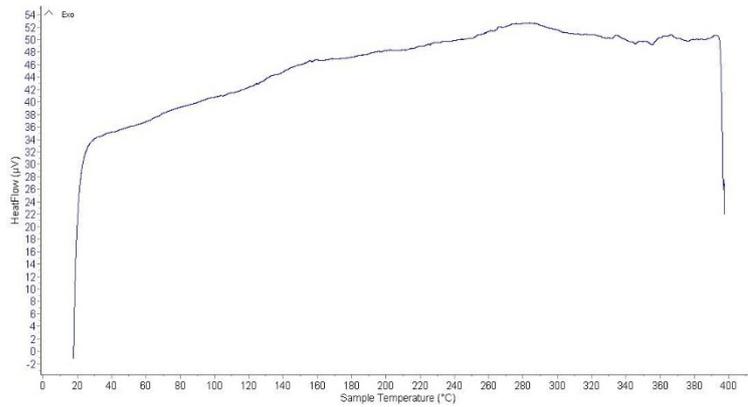


Figure 7: 1% Ag element.

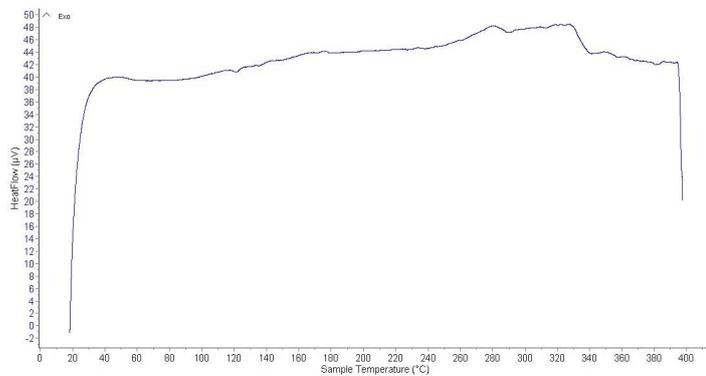


Figure 8: 2% Ag element.



Figure 9: 3% Ag element.

IV. SME

Shape Memory Effect (SME) is considered an important property of SMAs. As mentioned previously in experimental work and according to Eq. (1), the shape memory effect is about 89.99% which in turn is affected on physical properties. The improvement in most properties of SMAs manufactured with different atomic ratios is due to the purity of the materials used, the manufacturing method, uniform mixing, and the annealing process.

4. CONCLUSIONS

- 1) The results of the method used to manufacture shape memory alloys (SMAs) showed that the produced alloys are homogeneous, have uniform silver distribution, high purity alloys and have a shape memory effect of (89.99%). This indicates the success of the manufacturing process by means of the casting process using a (VAR) furnace.
- 2) The examinations conducted on the four samples with different percentage of the silver element (0, 1, 2 and 3 at. % Ag) by FESEM examination showed that the homogeneous silver distribution in NiTi matrix is due, in addition to the above parameters, to the emergence of martensite phase, austenite phase and some impurities as well.
- 3) The results of the XRD examination showed the emergence of martensite phase (Ti 002) with monoclinic structure upon cooling phase, Austenite phase (Ti₂Ni) with cubic structure upon heating and undesirable phase (Ni₄Ti₃) which is created as a result of the phase transformations well as some impurities resulting from the manufacturing process. This examination shows also the increasing appearance of peaks with the increase of the amount of silver.
- 4) The results of the DSC examination reveal the starting and finishing phase transformation temperatures (As, Af, Ms, Mf) for all alloys, which indicates the thermal transition from the austenite phase (B2) to the martensite phase (B19).

Acknowledgment

I extend my thanks, dedication and respect to the esteemed supervisors Assist. Prof. Dr. Khansaa Dawood Salman and Lec. Dr. Ahmed Abdulqader Hussein for their assistance in carrying out this research.

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