

## Effect of Annealing on Some Properties of $Zn_2SnO_4$ Thin Films Prepared by PLD Technique

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### Abstract

In this paper, the investigation of structural and optical properties of  $Zn_2SnO_4$  thin films were studied. The films are performed on glass substrates by pulsed laser technique (PLD) using laser Nd: YAG at wavelength of 1064 nm with 800 mj laser energy using repetition rate of 6 Hz. and average 400 laser pulses at room temperature and annealing by tubular quartz furnace at temperature (573,773) K for 2 hours with air. XRD measurements showed that the structure for all samples is polycrystalline with a cubic nanostructure. Surface morphology was studied using scanning electron microscopy SEM and atomic force microscopy AFM. After annealing, the roughness of the surface and the mean grain size were increased. Optical properties as a function to wavelength in the range (300-1100 nm) have been studied. Absorption spectra of  $Zn_2SnO_4$  thin films showed that absorption decreases with increasing annealing temperature. Direct energy gap for a  $Zn_2SnO_4$  thin film was increases with increasing temperature for all samples due to crystal growth. The optical properties such as extinction coefficient, refractive index, and dielectric constant were also studied.

**Keywords:** annealing;  $Zn_2SnO_4$  ; PLD technique.

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## تأثير التلدين على بعض خواص اغشية $Zn_2SnO_4$ الرقيقة المحضرة بتقنية

### PLD

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### المخلص

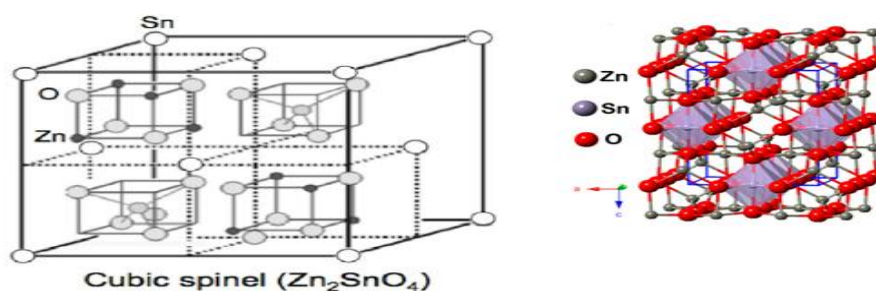
تم في هذا البحث دراسة الخصائص التركيبية والبصرية لأغشية  $Zn_2SnO_4$  الرقيقة والتي رسبت على قواعد زجاجية بواسطة تقنية ترسيب الليزر النبضي (PLD) باستخدام ليزر Nd: YAG النبضي عند طول موجة (1064 نانومتر) باستخدام طاقة ليزر (800 ملي جول). ومعدل تكرار قدره (6 Hz). باستخدام متوسط 400 نبضة ليزر. عند درجة حرارة الغرفة ثم التلدين بواسطة فرن الكوارتز الأنوبي عند درجة حرارة (573، 773) كلفن لمدة ساعتين بوجود الهواء. أظهرت قياسات (XRDs) أن البنية البلورية لجميع العينات هي متعددة التبلور ذو تركيب مكعبي نانوي. تمت دراسة مورفولوجيا السطح باستخدام المجهر الإلكتروني الماسح (SEM) ومجهر القوة الذرية (AFM). بعد التلدين، تزداد خشونة السطح ومتوسط الحجم الحبيبي. تمت دراسة الخواص البصرية كدالة الى الطول الموجي في المدى (300-1100 نانومتر). وأظهرت أطياف الامتصاص لأغشية  $Zn_2SnO_4$  أن الامتصاصية تتخفض مع زيادة درجة حرارة التلدين. كما ان فجوة الطاقة المباشرة للأغشية تزداد مع زيادة درجة الحرارة لجميع العينات بسبب النمو البلوري. تمت دراسة الخصائص البصرية مثل معامل الخمود، معامل الانكسار ، وثابت العزل الكهربائي.

الكلمات الدالة: التلدين،  $Zn_2SnO_4$ ، بتقنية PLD.

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## 1. Introduction:

Zinc stannate  $Zn_2SnO_4$ , commonly known as zinc tin oxide ZTO, has been confirmed for high electron mobility ( $10-15 \text{ cm}^2 / \text{V.s}$ ). High electrical conductivity,  $Zn_2SnO_4$  is a n-type transparent conductive oxide with a wide band gap of 3.7 eV [1]. It has wonderful visual optical properties that can be used in multiple fields such as solar cells, sensors to detect moisture and combustible gases. Fig 1. Crystalline structures of  $Zn_2SnO_4$  (cubic spinel) [2]. Pulsed laser deposition is defined as a promising method for depositing metal thin films. The thickness of the films can be controlled by controlling the number of pulses [3]. The idea of PLD is simple. The laser pulse beam is concentrated on the outer surface of the target material. Where the laser is absorbed and leads to rapid evaporation of the target material [4]. Evaporated materials include highly ionized species. If the discharge is performed in a vacuum, it shows itself as a glowing plasma column directly opposite the target surface. In our current research, we study the structural and optical properties of  $Zn_2SnO_4$  thin films grown on a glass slide produced by the PLD [5].



**Fig. 1:** Crystalline structures of zinc stannate  $Zn_2SnO_4$  (cubic spinel) [2].

## 2. Theory:

Zinc oxide ZnO and tin oxide  $SnO_2$  is the most promising candidate for the development of transparent conductive material.  $SnO_2$  is an rutile tetragonal structure with oxygen deficient n- type degenerate semiconductor with wide band gap of 3.6 eV. Its high optical transparency and electrical conductivity leads to very appealing applications in spintronics device [6]. ZnO in view of its high transmission over a wide spectral range including the useful UV-vis region and other interesting characteristics such as low toxicity, relatively low cost, and stability in reductive chemical environments [7].  $SnO_2$  and ZnO thin films have attracted significant interest recently for use in optoelectronic application such as solar cells, flat panel displays, photonic devices, laser diodes and gas sensors. In the present study, Nanoparticle powders of zinc stannate in date spinel  $Zn_2SnO_4$  were prepared by mixing stoichiometric amounts of tin

oxide SnO<sub>2</sub> and zinc oxide ZnO, in the same molar ratio, (ZnO:SnO<sub>2</sub> = 1:1).The zinc stannate Zn<sub>2</sub>SnO<sub>4</sub> films have the advantages of both ZnO and SnO<sub>2</sub>,and are promising for solar cell and sensor applications [8].

### 3. Experimental Work:

Tin oxide Nanoparticles SnO<sub>2</sub> purity (99.98 %) powder by Sky Spring Nanoparticles, Inc. 2935 Westhollow Dr., Houston, Tx 77082. and zinc oxide ZnO Nano Powder, with purity (99.9%) by Nanjing Nano Technology. Prepared Zn<sub>2</sub>SnO<sub>4</sub> by mixing stoichiometric amounts of tin oxide SnO<sub>2</sub> and zinc oxide ZnO, in the molar ratio, (ZnO:SnO<sub>2</sub> = 1:1) and ball milled for 1 hrs. to use it to make target as a disk of (1.5) cm diameter and (0.3) cm thickness using hydraulic piston type (SPECAC), under pressure of (6) tons for (15) minutes. finally the pellets were sintered in air at (1000 C°) for (2) hrs.

**PLD and deposition of thin film:** We used glass slides (2.5 x 7.5 cm) that were cleaned with liquid soap. Then immersed in distilled water using an ultrasound device for 15 minutes to remove all dust and dirt. Dry it with a thin strip of cotton to be ready for deposition at room temperature. Use PLD technique using the laser Nd: YAG with ( $\lambda = 1064$  nm) at power (800 mJ), under the vacuum ( $3 \times 10^{-3}$  mp) using the double- stage rotary pump. Repeating frequency (6Hz) for laser pulse (400) is incident on the target surface with an angle of (45°). A set of PLD system is given in Fig. 2. The distance between the target and the laser (10 cm) was determined ,and between the target and the glass slide (2 cm).

**Annealing process:** The annealing process was performed by a tubular quartz furnace Fig. 3. at a temperature of (573,773) K° for 2 hours with air.

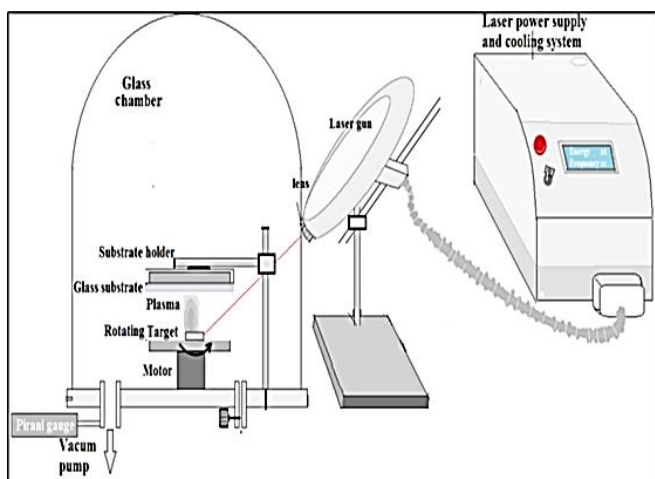


Fig. 2: Schematic of the PLD system [9]



Fig. 3: tubular quartz furnace.

**X-Ray Diffraction:** The crystalline structure of the  $Zn_2SnO_4$  thin films, which were deposited on glass slides by PLD method at RT, and ( $573,773 K^\circ$ ) temperatures, was studied using x-ray diffraction techniques using (Bruker-D2 phaser vorsicht rontgenstrahl ung caution x-rays) device. This device registers the intensity as a function of the Bragg's angle [10]:

$$n \lambda = 2 d_{hkl} \sin \theta \quad (1)$$

Where  $\theta$  is the diffraction angle and  $\lambda$  is the XRD wavelength used. The grain size of the granularity  $G.S$  of the crystals, which is important in the study of X-ray spectrum characteristics, can be estimated by full width at half-maximum (FWHM) calculated by Scherrer equation [11]:

$$G.S = \frac{0.9 \lambda}{FWHM \cos \theta} \quad (2)$$

Where  $\theta$  is the Bragg's angle of the XRD Peak.

**Measuring optical properties:** The optical properties of thin films are affected by crystalline structure, thickness and type of materials used. The optical properties have been investigated using UV-visible spectral spectrometer (SP-8001) in the range (300-1100) nm. The output data from the absorption, wavelength, and transmitter are used in a computer program to measure all optical constants. The value of the energy gap ( $E_g$ ) is graphically estimated by equation [12]

$$\alpha h\nu = B' (h\nu - E_g^{opt.})^{\frac{1}{2}} \quad (3)$$

Were  $E_g^{opt.}$  = is the optical energy gap ;  $h$  Blank's constant ;  $\nu$  is the frequency of light;  $B'$  constant depends on the type of material.

When measuring absorption and transmittance spectra, we can know the behavior of the refractive index spectra  $n$  and extinction  $k$  of the ZTO [13].

$$k = \frac{\alpha \lambda}{4\pi} \quad (4)$$

$$n = \left( \frac{4R}{(R-1)^2} - k^2 \right)^{1/2} - \frac{(R+1)}{(R-1)} \quad (5)$$

where  $R$  is the reflectance. The real and imaginary part of dielectric constant can be calculated by using the following equations [14]:

$$\epsilon_r = n^2 - k^2 \quad (6)$$

$$\varepsilon_i = 2nk \quad (7)$$

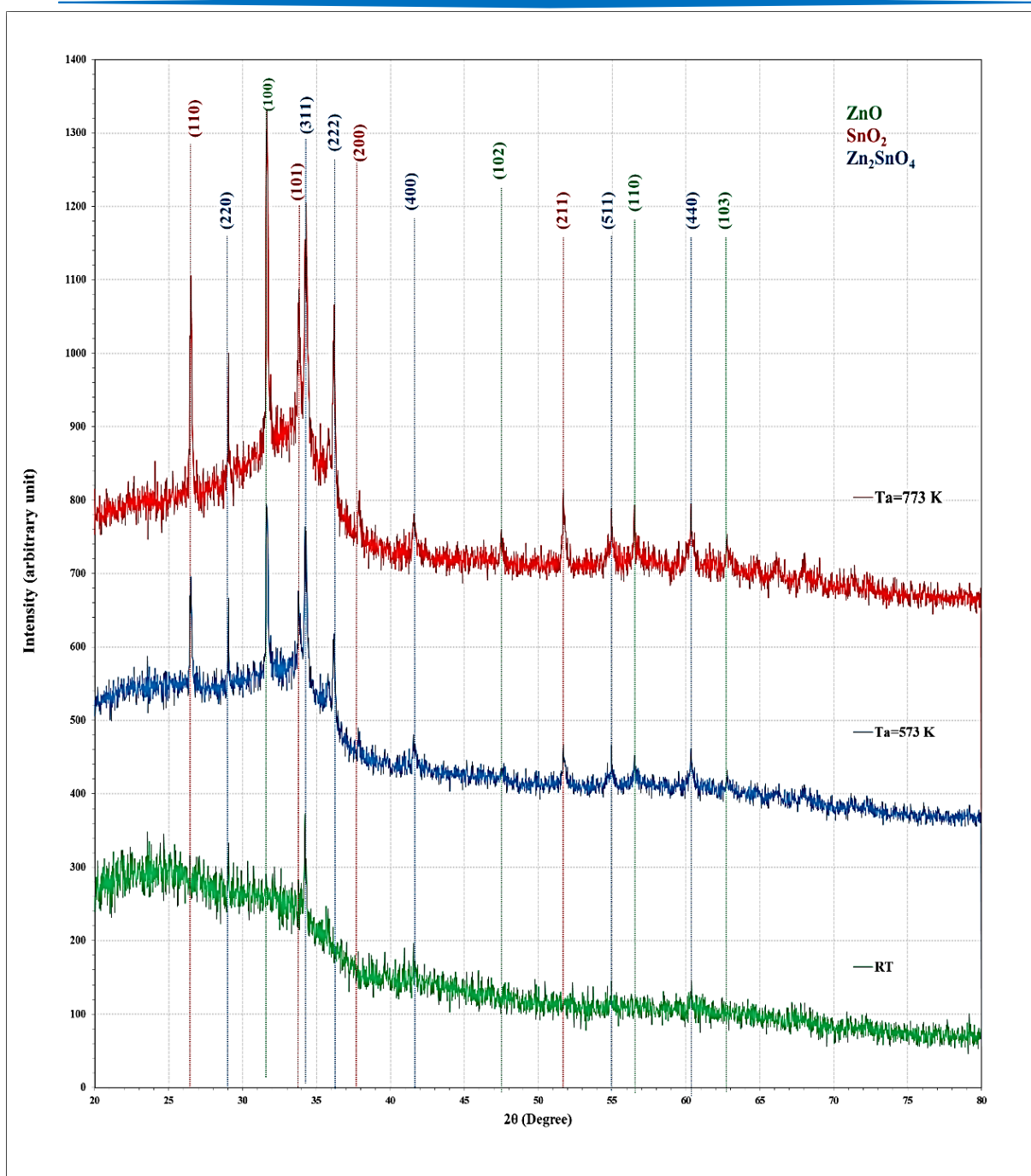
## 4. Results and Calculations:

**4.1 Structure measurement:** Fig. 4 shows the X-ray diffraction for  $Zn_2SnO_4$  thin films. We can observe that the deposited film has polycrystalline structure, XRD diffraction results at room temperature have two peak located at  $2\theta$  values (29.0499, 34.2162) corresponding to (220) and (311) direction respect to  $Zn_2SnO_4$  crystals. and with annealing at (573,773) K°, It shows several Six diffraction peaks at  $2\theta$  values of (29.0499°, 34.2162°, 36.1758°, 41.5558°, 54.9881° and 60.3325°).

These were assigned to the diffraction lines produced by (220), (311), (222), (400), (511) and (440) of the (cubic spinel) structure of  $Zn_2SnO_4$  confirmed by standard data. We can see many other peaks are detected in the XRD pattern for the hexagonal ZnO at  $2\theta$  values of (31.6152°, 47.6128°, 56.5558° and 62.791°) and tetragonal  $SnO_2$  at  $2\theta$  values of (26.5202°, 33.7886°, 37.8504° and 51.7102°).

The FWHM decrease with increasing annealing temperature indicate on increasing particle size. The intensity of the peaks increases as the temperature of the annealing increases, which means that the crystallization of the film improves as the temperature of the annealing increases.

The annealing process reduces crystalline defects and reduce optical dispersion. Table 1 shows the experiment and the standard peaks from (JCPDS Joint Committee on Powder Diffraction Standards - (ICDD) International Centre for Diffraction Data) cards number (24-1470, 96-210-4744 and 96-901-1663) Returns to  $Zn_2SnO_4$ ,  $SnO_2$  and ZnO Respectively.



**Fig. 4:** X-ray diffraction patterns of Zn<sub>2</sub>SnO<sub>4</sub> films deposited at RT and different annealing temperatures (573,773) K°.

**Table 1:** Structural parameters: Inter-planar spacing, crystalline size of Zn<sub>2</sub>SnO<sub>4</sub> films deposited at RT and annealing at (573,773) K°.

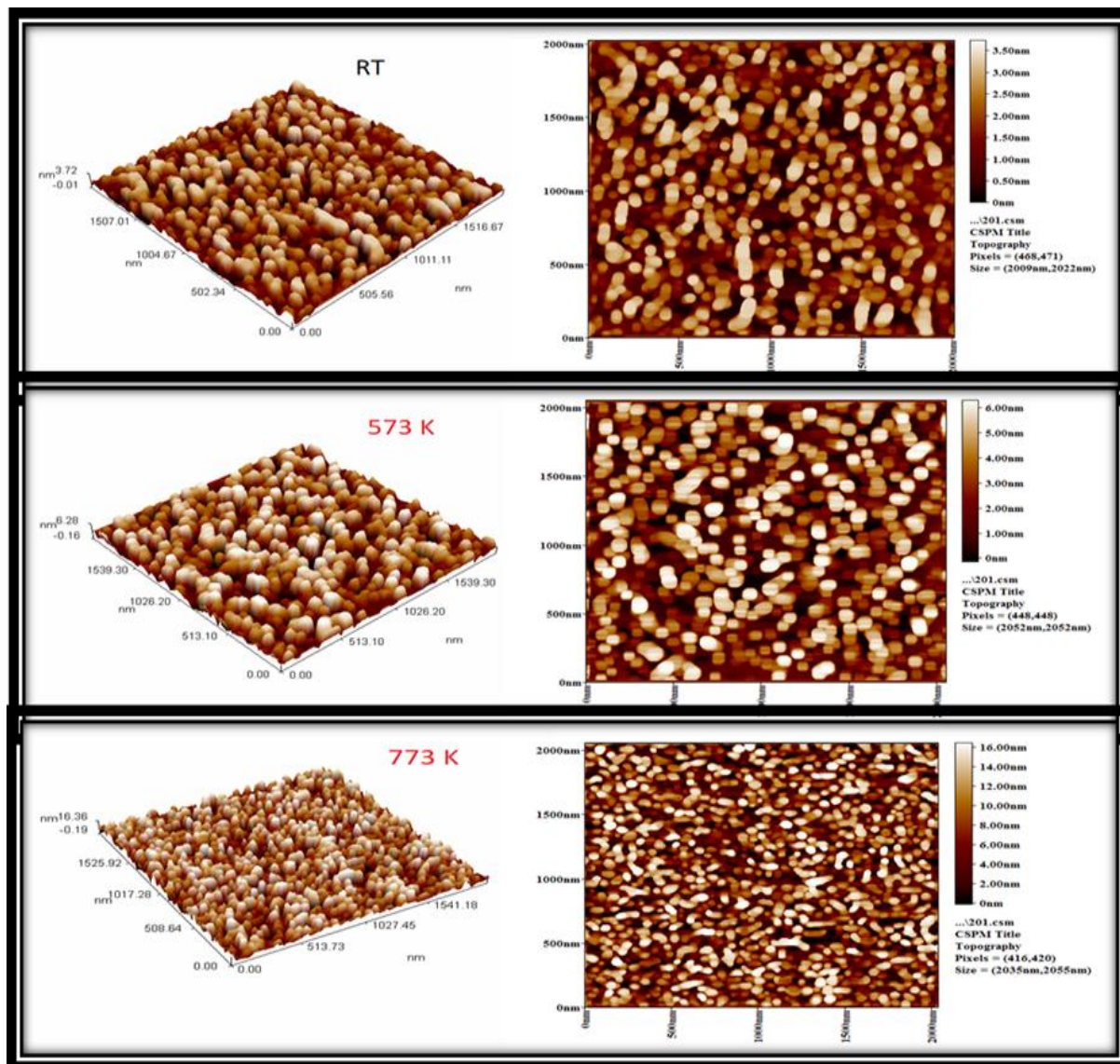
T <sub>a</sub> (K)	2θ (Deg.)	FWHM (Deg.)	d <sub>hkl</sub> Exp.(Å)	G.S (nm)	Hkl	d <sub>hkl</sub> Std.(Å)	Phase	Card No.
RT	29.0499	0.2494	3.0714	32.9	(220)	3.0621	Zn <sub>2</sub> SnO <sub>4</sub>	24-1470
	34.2162	0.3207	2.6185	25.9	(311)	2.6130	Zn <sub>2</sub> SnO <sub>4</sub>	24-1470
573	26.5202	0.2138	3.3583	38.2	(110)	3.3498	SnO <sub>2</sub>	96-210-4744
	29.0499	0.2138	3.0714	38.4	(220)	3.0621	Zn <sub>2</sub> SnO <sub>4</sub>	24-1470
	31.6152	0.1781	2.8277	46.4	(100)	2.8137	Hex. ZnO	96-901-1663
	33.7886	0.2137	2.6507	38.9	(101)	2.6439	SnO <sub>2</sub>	96-210-4744
	34.2162	0.2850	2.6185	29.2	(311)	2.6130	Zn <sub>2</sub> SnO <sub>4</sub>	24-1470
	36.1758	0.2494	2.4810	33.5	(222)	2.4991	Zn <sub>2</sub> SnO <sub>4</sub>	24-1470
	37.8504	0.2494	2.3750	33.7	(200)	2.3686	SnO <sub>2</sub>	96-210-4744
	41.5558	0.3563	2.1714	23.9	(400)	2.1651	Zn <sub>2</sub> SnO <sub>4</sub>	24-1470
	47.6128	0.3563	1.9083	24.4	(102)	1.9110	Hex. ZnO	96-901-1665
	51.7102	0.3563	1.7664	24.8	(211)	1.7642	SnO <sub>2</sub>	96-210-4744
	54.9881	0.4988	1.6686	18.0	(511)	1.6647	Zn <sub>2</sub> SnO <sub>4</sub>	24-1470
	56.5558	0.6057	1.6260	14.9	(110)	1.6245	Hex. ZnO	96-901-1663
	60.3325	0.4988	1.5329	18.4	(440)	1.5304	Zn <sub>2</sub> SnO <sub>4</sub>	24-1470
62.7910	0.6057	1.4787	15.4	(103)	1.4772	Hex. ZnO	96-901-1663	
773	26.5202	0.2100	3.3583	38.9	(110)	3.3498	SnO <sub>2</sub>	96-210-4744
	29.0499	0.1581	3.0714	51.9	(220)	3.0621	Zn <sub>2</sub> SnO <sub>4</sub>	24-1470
	31.6508	0.1425	2.8246	58.0	(100)	2.8137	Hex. ZnO	96-901-1663
	33.8242	0.2010	2.6480	41.3	(101)	2.6439	SnO <sub>2</sub>	96-210-4744
	34.2874	0.2494	2.6132	33.4	(311)	2.6130	Zn <sub>2</sub> SnO <sub>4</sub>	24-1470
	36.1758	0.2138	2.4810	39.1	(222)	2.4991	Zn <sub>2</sub> SnO <sub>4</sub>	24-1470
	37.9216	0.2494	2.3707	33.7	(200)	2.3686	SnO <sub>2</sub>	96-210-4744
	41.5914	0.3206	2.1696	26.5	(400)	2.1651	Zn <sub>2</sub> SnO <sub>4</sub>	24-1470
	47.5059	0.2851	1.9124	30.5	(102)	1.9110	Hex. ZnO	96-901-1665
	51.6746	0.2851	1.7675	31.0	(211)	1.7642	SnO <sub>2</sub>	96-210-4744
	54.9525	0.3919	1.6696	22.9	(511)	1.6749	Zn <sub>2</sub> SnO <sub>4</sub>	24-1470
	56.5558	0.2425	1.6260	37.2	(110)	1.6245	Hex. ZnO	96-901-1663
	60.3682	0.2138	1.5321	43.0	(440)	1.5304	Zn <sub>2</sub> SnO <sub>4</sub>	24-1470
62.7910	0.2851	1.4787	32.7	(103)	1.4772	Hex. ZnO	96-901-1663	

## 4.2 Morphology Analysis :

**4.2.1. Atomic Force Microscope (AFM):** Fig.5 shows the AFM images and their granularity accumulation distribution for Zn<sub>2</sub>SnO<sub>4</sub> thin films deposited on glass slides by PLD at (RT) and annealing at (573,773 K°). This Figure and Table 2 illustrate that the Average Diameter, roughness Average and RMS are increased with increased annealing temperature. This may



be due to the bigger clusters formed by the coalescence of two or more grains and decrease or disappearance the grain boundaries. This result is identical with [15].



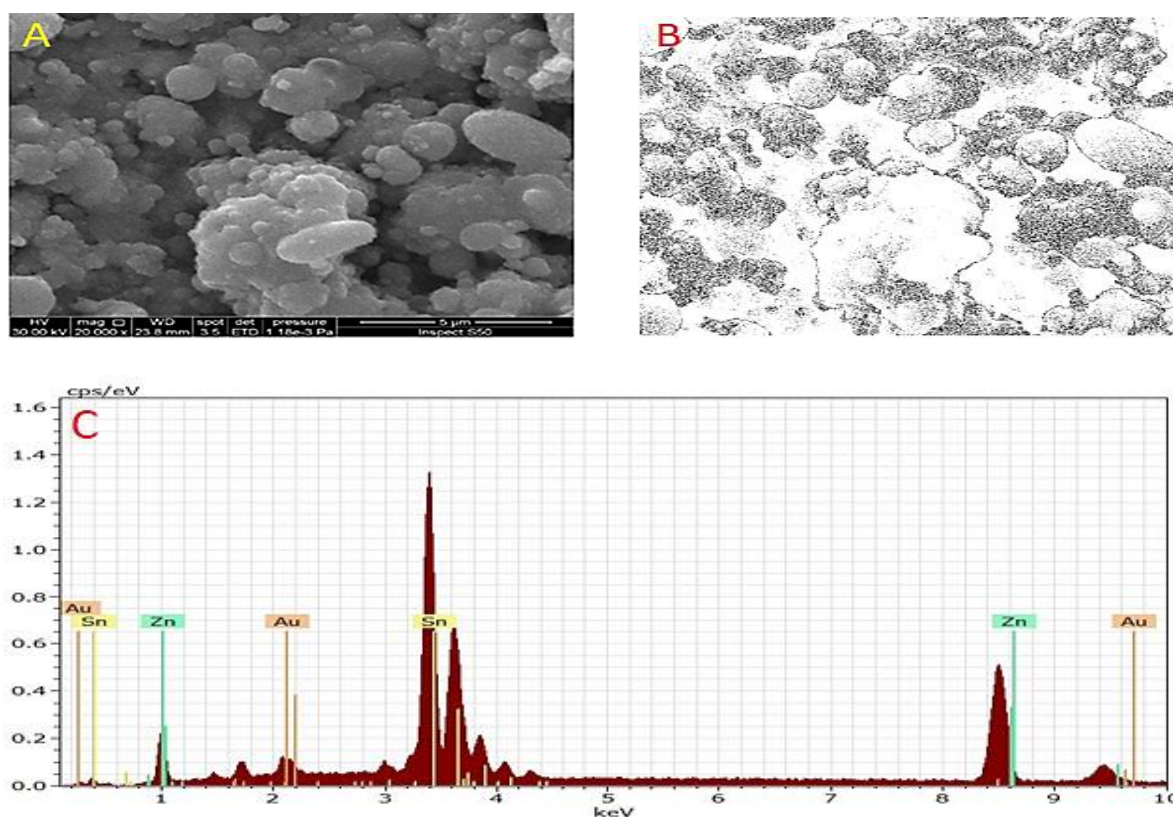
**Fig. 5:** AFM images and their granulate accumulation for  $Zn_2SnO_4$  thin films deposited at RT and annealing at (573,773) K°.

**Table 2:** AFM parameters (Average Diameter, roughness Average and RMS) for  $Zn_2SnO_4$  films deposited at RT and annealing at (573,773) K°.

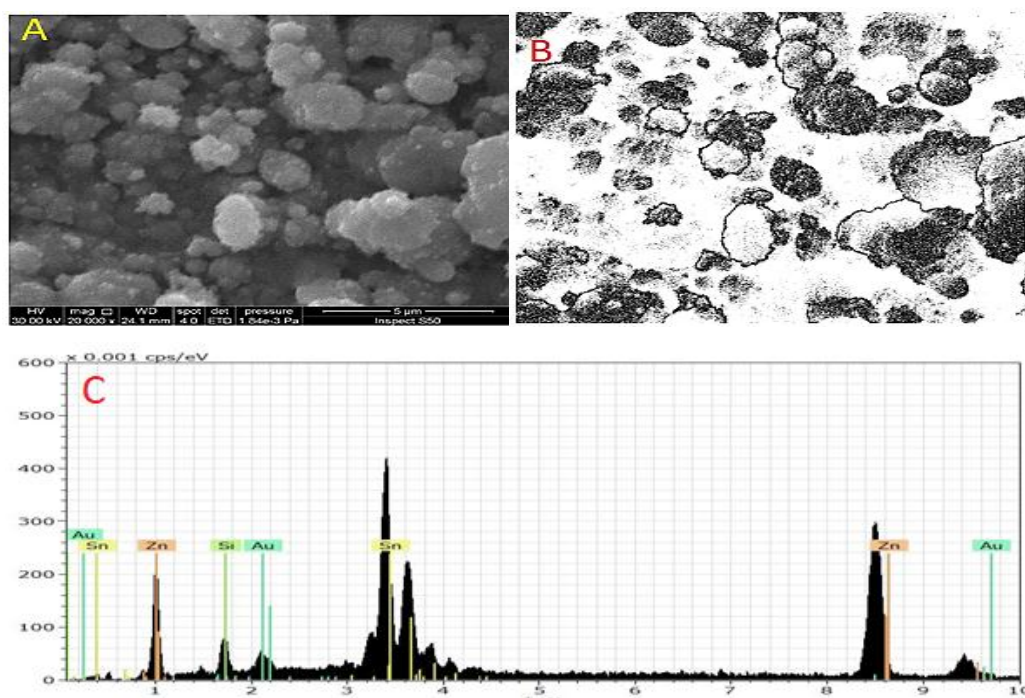
Temperature (K°)	Average Diameter (nm)	roughness Average (nm)	Root Mean Square (nm)
RT	63.96	0.832	3.17
573	69.65	1.61	6.44
773	80.31	4.14	16.5

**4.2.2. scanning electron microscopy (SEM):** Fig. 6a,7a shows the SEM images of  $Zn_2SnO_4$  films deposited on the glass substrates by the PLD at RT and its temperature 773 K°. Fig. 6b,7b shows the SEM analysis images with the IMAGEJ program. The surface of all thin films is known as a cluster of granules known as clusters, which represents homogeneous surface morphology of thin films. The  $Zn_2SnO_4$  thin film has a perfectly uniform surface with no holes. With a wide distribution of particles of about 20-50 nm, which agrees with the XRD result. With annealing, the average size of aggregated particles increases obviously. This result may be attributed to higher atom mobility with the increase in temperature which causes more effective recrystallization and grain growth of the films that result in larger grains. This result was in agreement with [16].

**4-2.3 EDAX Study:** The EDAX pattern shown in Fig. (6-c), (7-c) confirms the presence of Zn and Sn. The observed Zn/Sn ratio for the films measured from the EDX measurement of the  $Zn_2SnO_4$  stoichiometry for both the sediment by PLD in RT and annealed at (773) K°.



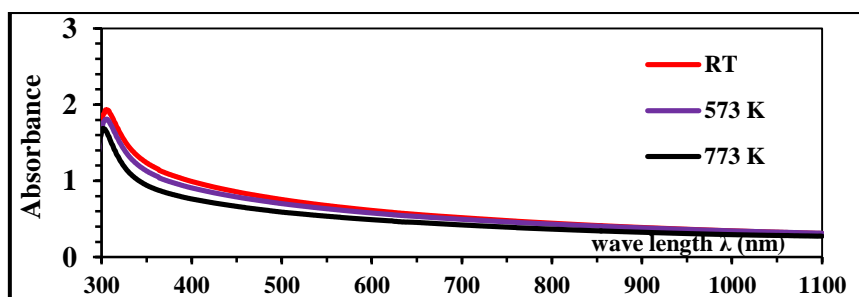
**Fig. 6:**  $Zn_2SnO_4$  thin films deposited at RT (a) SEM image, (b) Analyze SEM image by IMAGEJ program, (C) Typical EDAX pattern.



**Fig. 7:**  $Zn_2SnO_4$  thin film annealed at (773) K° temperature (a) SEM image, (b) Analyze SEM image by IMAGEJ program, (C) Typical EDAX pattern.

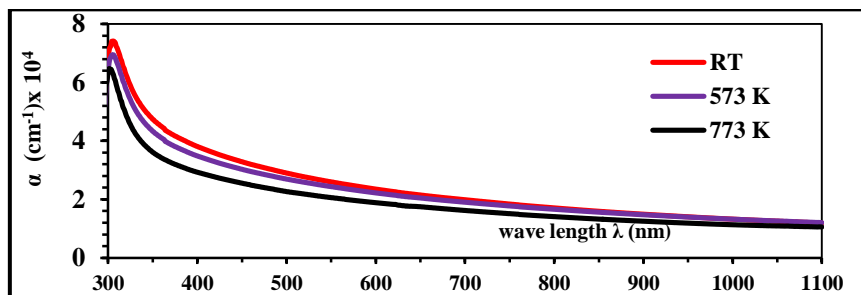
### 4.3 Optical measurement :

**4.3.1 Absorbance spectrum:** Fig. 8 shows the absorption spectrum as a function of the wavelength for a range (300-1100) nm. It can be seen that when the temperature of the annealing increases, as a result of increase the crystallization, the amount of absorption decreases. The reason is the decay and disappearance of donor levels (i.e localized states ) within forbidden energy gap and conduction band. This leads to an increase in the value of  $E_g$ . In the near UV-visible light region, a strong photo-absorption at a wavelength of 330 nm is presented. This result is in agreement with [2].



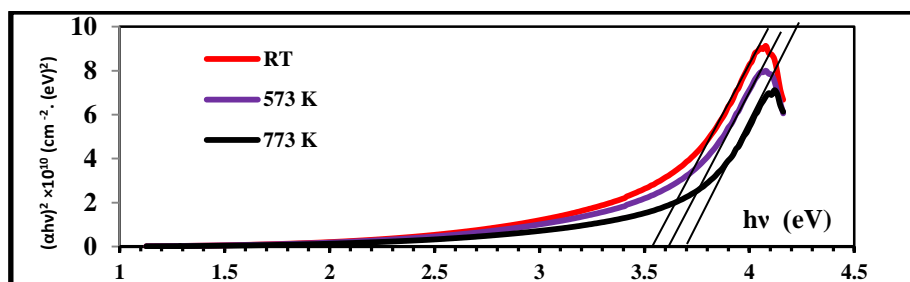
**Fig. 8:** shows absorbance for  $Zn_2SnO_4$  films deposited at RT and annealing at (573,773) K°.

**4.3.2 Absorption Coefficient ( $\alpha$ ):** Fig. 9 show the optical absorption coefficient  $\alpha$  as a function of wavelength ( $\lambda$ ) for  $Zn_2SnO_4$  thin films. we can see from this Figure that the values of  $\alpha$  becomes higher ( $\alpha > 10^4$ )  $cm^{-1}$ , this supports to expect a direct electronic transition occurs in these regions. also we found that the  $\alpha$  of the  $Zn_2SnO_4$  films has a strong absorption coefficient at the short wavelength region (high energies). On the other hand, it is found that the value of  $\alpha$  decreases with increasing of annealing temperature due to increase energy gap  $E_g$ . The  $\alpha$  values are approximately equal to the reported values by [2].



**Fig. 9:** shows Absorption Coefficient for  $Zn_2SnO_4$  films deposited at RT and annealing at (573,773) k

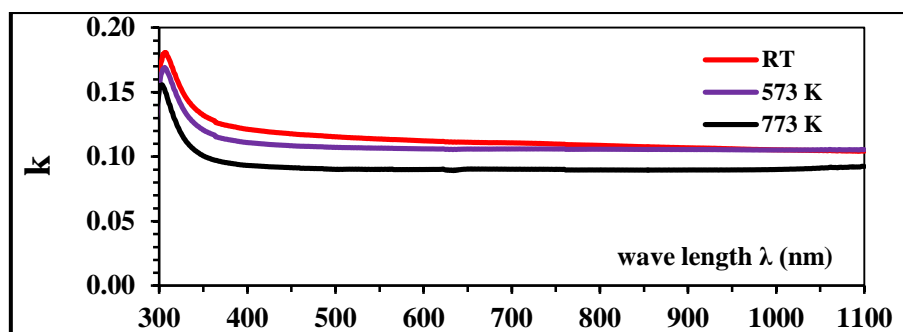
**4-3-3 The Optical Energy Gap ( $E_g$ ):** The values of the optical energy gap  $E_g$  for nanoparticles  $Zn_2SnO_4$  thin films are determined using eq (3). A plot of relation  $(\alpha hv)^2$  versus photon energy ( $hv$ ) and the choice of the desired linear section. From Fig. 10, we can see that the energy gap was increased from (3.55-3.7) eV, as a result of the increasing of the annealing temperature, This is due to the growth of grain size and the reduction of the number of grain boundaries. Increase  $E_g$  as a result of reducing the amount of absorption, this may lead to improved the crystal structures, which leads to the reduction of defects (tail state) in the forbidden gap and this leads to an increase in the  $E_g$ . The values of the approximate absorption coefficient with the values reported by [17].



**Fig. 10:**  $(\alpha hv)^2$  as function of energy photon for nanostructure  $Zn_2SnO_4$  thin films deposited at RT and annealing at (573,773) K°.

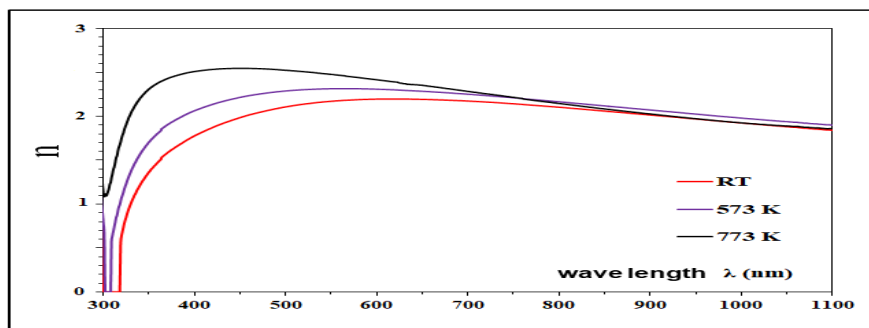
#### 4.3.4 Optical Constants:

**4.3.4.a Extinction Coefficient ( $K$ ):** According to the formula. (4) The behavior of the extinction coefficient ( $K$ ) is similar to the behavior of the absorption coefficient ( $\alpha$ ). Fig. 11 shows the extinction coefficient ( $K$ ) as a function of the wavelength ( $\lambda$ ) of  $Zn_2SnO_4$  films at RT and (573,773) K°. We can see from Fig. 11 and Table. 3 that the extinction coefficient ( $K$ ) decreases generally with increased temperatures. This result is consistent with [18].



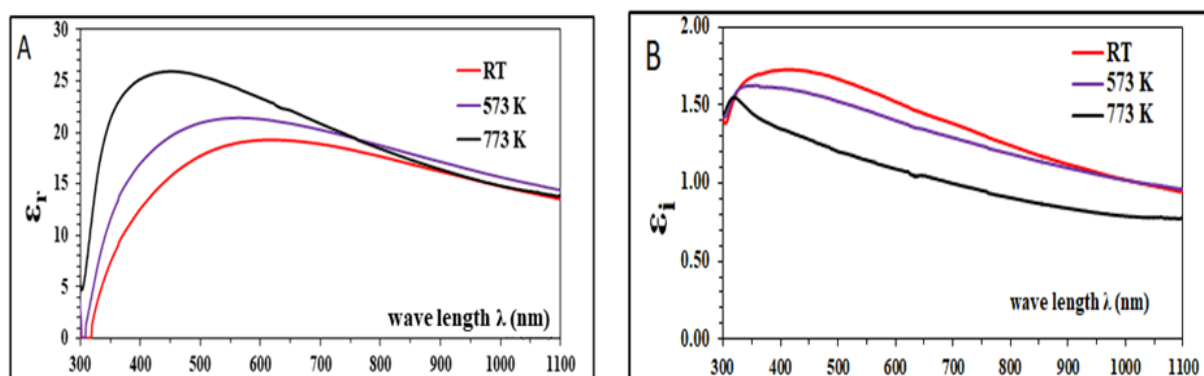
**Fig. 11:** Extinction coefficient  $K$  with wavelength for nanostructure  $Zn_2SnO_4$  thin films deposited at RT and annealing at (573,773) K°.

**4.3.4.b Refractive Index ( $n$ ):** From Fig. 12 we see that the value of Refractive index ( $n$ ) for all films is approximately limited by (2.2.5). we can show from these figures that ( $n$ ) increases slightly with increasing of annealing temperature and the reason for this, first the growth of grain size which causes increases of Surface roughness which means increases of the reflection where the refractive index depends on it. Secondly, reduced porosity and increased density of the thin films material due to annealing which in turn decreases the light velocity thereby causing a change in the refractive index This is consistent with [19].



**Fig. 12:** The Variation of The Refractive Index  $n$  With Wavelength for  $Zn_2SnO_4$  thin films deposited at RT and annealing at (573,773) K°.

**4.3.5 Dielectric Constant:** The electrical dielectric Constant was studied in real and imaginary parts as a function of wavelength. Fig. 13 (a) shows the relationship between the real portion of the dielectric constant versus the wavelength where we see that all samples behave like refractive index ( $n$ ) samples due to the small value of ( $k^2$ ) compared to ( $n^2$ ) based on eq. (6) Fig. 13 (b) shows the relationship between the imaginary portion of the dielectric constant versus the wavelength. We found that all samples behaves like extinction coefficients ( $k$ ) based on equation (7). This means that the real part increases with increasing of annealing temperature The value of the imaginary part decreases as the temperature rises. It can be observed that the values of the real portion of the dielectric constant decrease with increasing wave lengths and actual values of the real part are greater than the values of the imaginary part as shown in Table 3 . These measurements correspond with [18].



**Fig. 13:** The Variation of The Dielectric Constants (a)-real part ( $\epsilon_r$ ) and (b)- imaginary ( $\epsilon_i$ ) parts With Wavelength of  $Zn_2SnO_4$  Films at RT and annealing at (573,773)  $K^\circ$ .

**Table 3:** Values of Absorption Coefficient  $\alpha$  ,Refractive index ( $n$ ), Extinction coefficient ( $k$ ), Real ( $\epsilon_r$ ) and Imaginary ( $\epsilon_i$ ) Dielectric Constant for  $Zn_2SnO_4$  films (at 500 nm wavelength) and The Optical Energy Gap ( $E_g$ ).

$T_a (K^\circ)$	$\alpha (cm^{-1})$	$K$	$n$	$\epsilon_r$	$\epsilon_i$	$E_g eV$
<b>RT</b>	28983	0.12	2.1	17.689	1.7167	3.55
<b>573</b>	26920	0.11	2.25	20.932	1.6355	3.6
<b>773</b>	22653	0.09	2.5	25.472	1.3659	3.7

## 5. Conclusion:

Structural and optical properties analysis show that pulsed laser deposition technique is a useful method for the deposition of  $Zn_2SnO_4$  thin films on glass substrates. Post-annealing process helps to improve the crystalline quality thin films. All the films show polycrystalline with cubic nanostructure. XRD, AFM and SEM indicate that the Average Diameter, roughness Average and RMS of the nanoparticles thin film increases with increasing the annealing temperature. direct energy band gap values increases with increasing annealing temperatures about 3.7 eV for the film annealed at 773 K for 2 hour.

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